

Haziot *et al.* Reply: The model presented by Zhou *et al.* in the preceding Comment [1] is inconsistent with the measurements we reported in Ref. [2], although it may be relevant to experiments involving much larger deformations.

The most striking inconsistency concerns the dependence of the shear modulus on the driving strain and the ^3He concentration. Zhou *et al.* ignore that the stiffness of ^4He crystals is highly sensitive to ^3He impurity concentration, an experimental fact that is well established since the work of Day and Beamish [3]. In our Letter [2], Fig. 2 shows that the transition temperature from a soft to a stiff state is shifted to a much lower temperature if the ^3He concentration is reduced from 300 to 0.4 ppb, and Fig. 5 shows a crystal—X4—where all ^3He impurities have been expelled in the adjacent liquid and which remains soft down to 20 mK even at very low driving strain. Since Zhou *et al.* claim that the stiffness and dissipation can be explained without the need for ^3He , let us compare their Fig. 1(c) with our measurements on X4 at 20 mK (Fig. 1). On a strain domain from 8×10^{-11} up to 1.5×10^{-7} , X4 shows a perfectly linear variation of the stress with no sign of any yield strain and a slope 5 times smaller than the intrinsic shear modulus (their initial slope). By replacing the effect of impurity binding by that of network pinning or Peierls barriers, Zhou *et al.* arrive to a striking contradiction with experiments.

Zhou *et al.* assume the existence of a yield strain 2.7×10^{-8} due to network pinning so that the transition temperature from soft to stiff changes drastically as a function of the applied strain [see Fig. 3(a) in Ref. [4]] and of the dislocation density (see Fig. 4 in Ref. [5]), again in contradiction with our experiments [2,6,7].

Zhou *et al.* show no results on the dependence on ^3He concentration but they consider that our results are “not sufficient evidence for the process of ^3He atoms binding to dislocations.” All our measurements are consistent with a stiffening due to the binding of ^3He impurities, especially the most recent ones [7].

From the imaginary part of the response to an ac drive [6], we have shown that dislocations move with no measurable dissipation in the “giant plasticity” regime. It is an experimental fact in contradiction with their model.

Zhou *et al.* assume that dislocations unbind from network nodes and that new dislocations are created beyond a yield strain 2.7×10^{-8} . However, in hcp crystals, pinning at nodes is normally much stronger than impurity pinning, so that dislocations should move only between the nodes in our case unless a very high strain is applied. How would these dislocations disappear during the sub-ms period of our ac strain, to give us the reversible, zero dissipation behavior we observe? In Ref. [6], we show that the dislocation density is stable, reproducible for a given crystal, and independent of time and of strain amplitude. We also find a dislocation displacement that

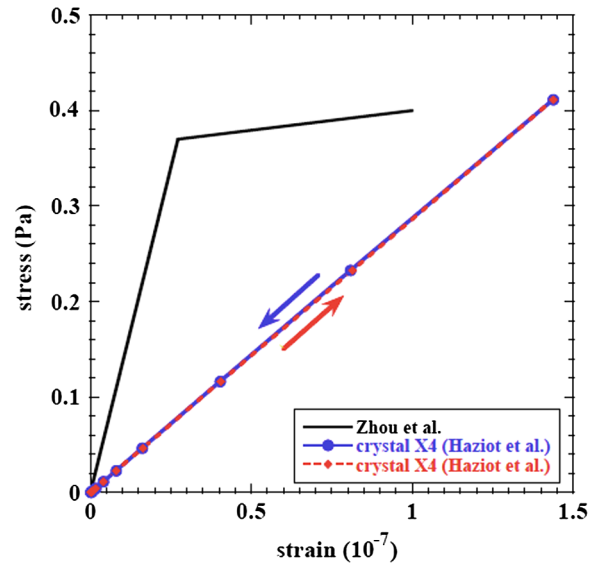


FIG. 1 (color online). The theory by Zhou *et al.* shows strong disagreement with our experiments. We observe [2] a linear and reversible behavior with highly reduced stiffness down to 8×10^{-11} strains, 300 times smaller than their yield strain (2.7×10^{-8}).

is much smaller than the interdislocation distance [6]. We work far below any threshold for dislocation creation, even when we measure the stress corresponding to break away from ^3He impurities. We have observed irreversible behavior that could be related to the creation of jogs on dislocations, but only at strain values that are orders of magnitude larger than in Ref. [2].

In their summary, Zhou *et al.* claim that our results “can be explained without requiring quantum tunneling or dissipationless motion of dislocations.” We carefully chose the title “Giant Plasticity of a Quantum Crystal” rather than, for example, “Quantum Plasticity in Solid Helium,” a description which has been used in the past. We explained in Ref. [2] that “an interesting question is whether dislocations move by quantum tunneling or by thermal activation. . .” because this is an open question.

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